

# Diffusion along dislocation cores in metals

How do atoms move in the dislocation core?

G. P. Purja Pun and Y. Mishin

Department of Physics and Astronomy, George Mason University,  
Fairfax, VA

Support: Air Force Office of Scientific Research  
(AFORS), Metallic Materials Program

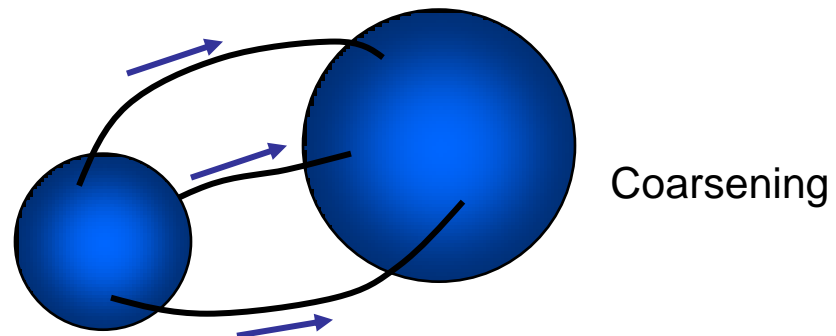
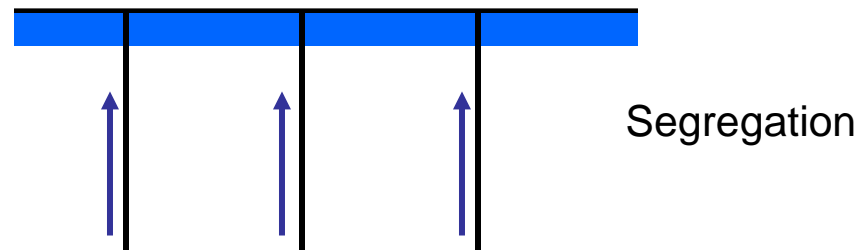
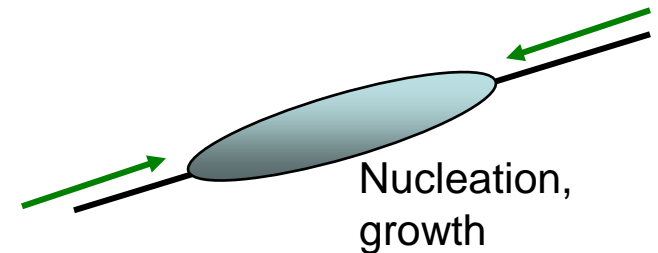
# Outline

- What do we know about dislocation (“pipe”) diffusion?
  - Experimental data
  - Theory and modeling
- Diffusion along a  $\frac{1}{2}\langle 110 \rangle$  screw dislocation in Al
  - Methodology
  - Results
  - Do we need point defects for dislocation diffusion?

# Dislocation diffusion in materials: Why is it important?

## Processes controlled/affected by dislocation diffusion:

- Precipitation and phase transformations
- Dynamic strain ageing
- Solute segregation
- Creep
- Coarsening
- Mechanical alloying
- Sintering
- Many others



## Measurements of dislocation diffusion

- **Direct** measurements: from concentration profiles.  $\Rightarrow P_d = \pi r_d^2 D_d$ 
  - Usually require radioactive isotopes
  - Based on simplified models: *a la* Fisher but with a regular arrangement of parallel dislocations (wall or network)
  - Interpretation of experimental data often problematic
  - Most of the direct measurements have been done in the 1960s-70s and summarized by Balluffi and Granato (1979). Few measurements in the 1980s-1990s; *very* few these days.
  - The most recent paper on dislocation self-diffusion in metals: Y. Shima et al. Mater. Trans. 43, 173 (2002), ultra-high-purity iron
  - SIMS can be used for impurity diffusion
- **Indirect methods:** from kinetics of processes
  - Internal friction
  - Dislocation climb
  - Dislocation loop shrinkage
  - Void shrinkage
  - Based on crude models with unknown parameters. Highly inaccurate

# Simulations of dislocation diffusion

- Mainly calculations of vacancy formation energies and jump barriers at 0 K. Low barriers - fast diffusion
- Identification of “high-diffusivity paths”
- Correlation factors are ignored. – Big mistake!
- Quasi-1D confinement may lead to strong correlation effects (Stark, late 1980s; Qin and Murch, 1993)
- MD simulations: diffusion coefficients of vacancies, not atoms
- Only vacancies were considered in almost all studies

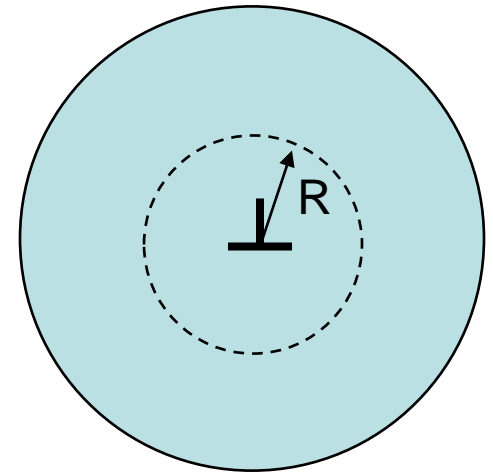
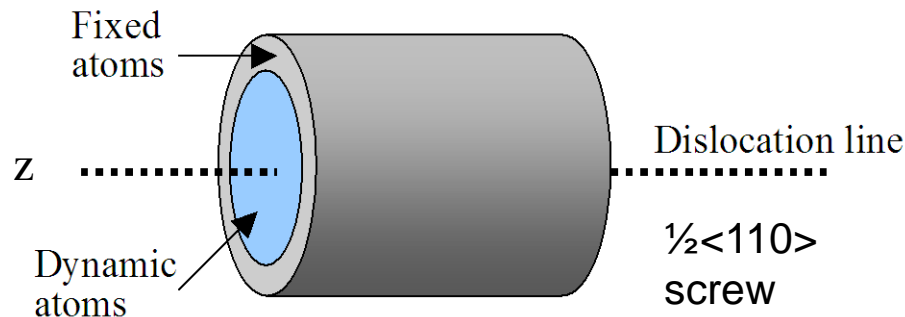
# What do we know about dislocation diffusion today?

- Not much...

- $D_d \gg D$ ;  $Q_d = (0.6-0.7)Q$ ; both depend on the dislocation Burgers vector and character (edge/screw)
- Diffusion is *believed* to be mediated by vacancies. The actual diffusion mechanisms remain unknown
- Analogy with GB diffusion suggest a variety of possible mechanisms

# Diffusion along a $\frac{1}{2}[110]$ screw dislocation in Al

- EAM potential for Al. Accurately reproduces  $c_{ij}$ ,  $\gamma_{SF}$ , point defects, diffusion, etc.
- Cylindrical block with dynamic and fixed atoms (7344 total)
- Dissociation into Shockley partials in agreement with experiment
- Introduce a single defect (vacancy or interstitial), or no defect
- Run MD for 30 nanoseconds at 750-1000 K ( $T_m = 1042$  K)
- MSD of **atoms** for 3-7 ns:  $\bar{D}(R) = \langle z^2 \rangle / 2t$
- Correction for equilibrium defect concentration

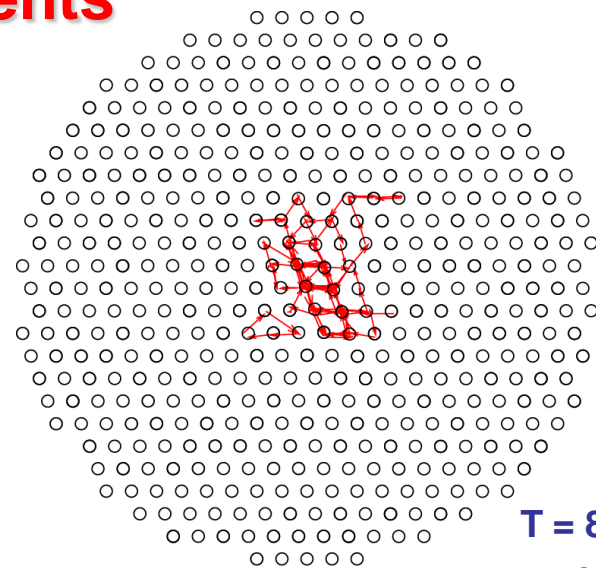
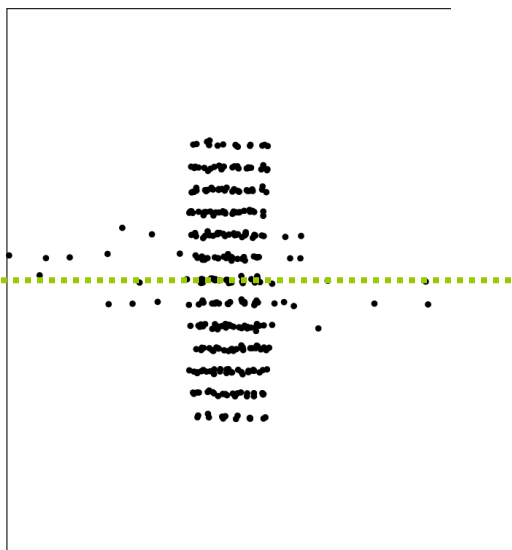


Output:  $\bar{D}(R)$

# Mean-squared displacements

1000 K

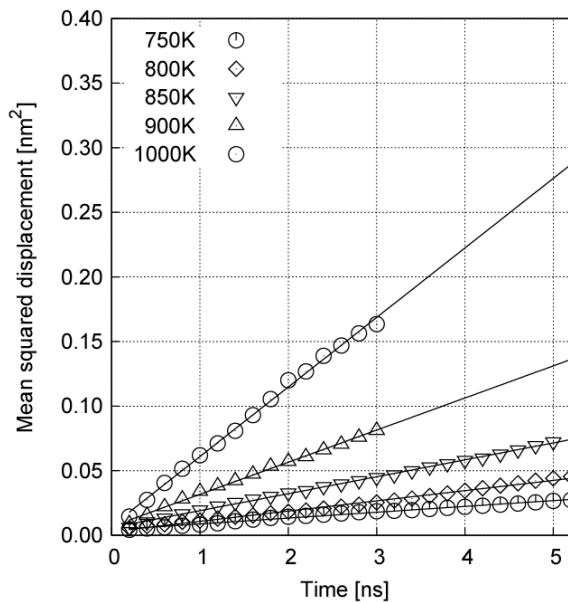
5 ns



T = 800 K

t = 3 ns

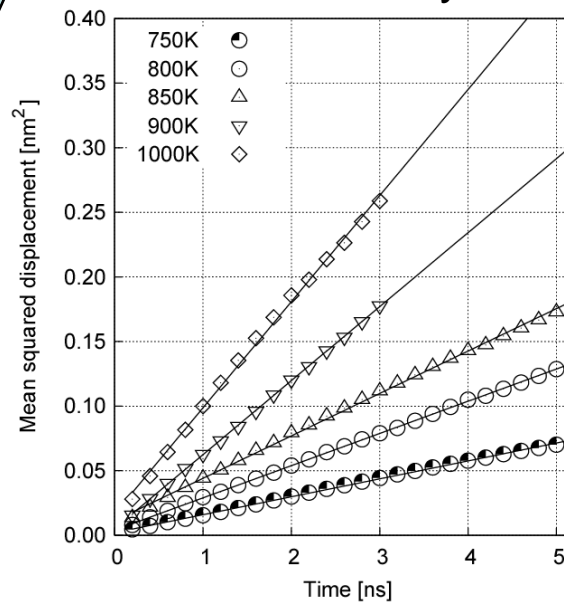
**Intrinsic!!!**



$\bar{D}(R)$

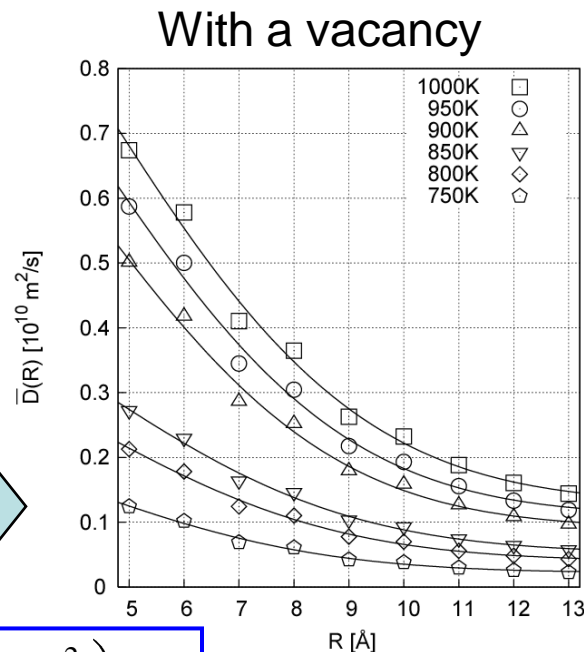
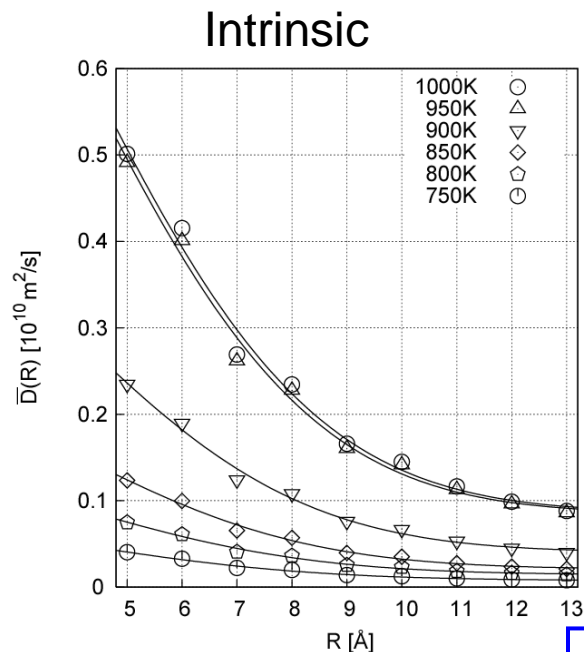
R=0.7 nm

**With a vacancy**





# How to find the dislocation diffusivity



$$\bar{D}(R) = A \exp\left(-\frac{R^2}{r_d^2}\right) + B$$

$$D_d^I \approx A/e; r_d$$

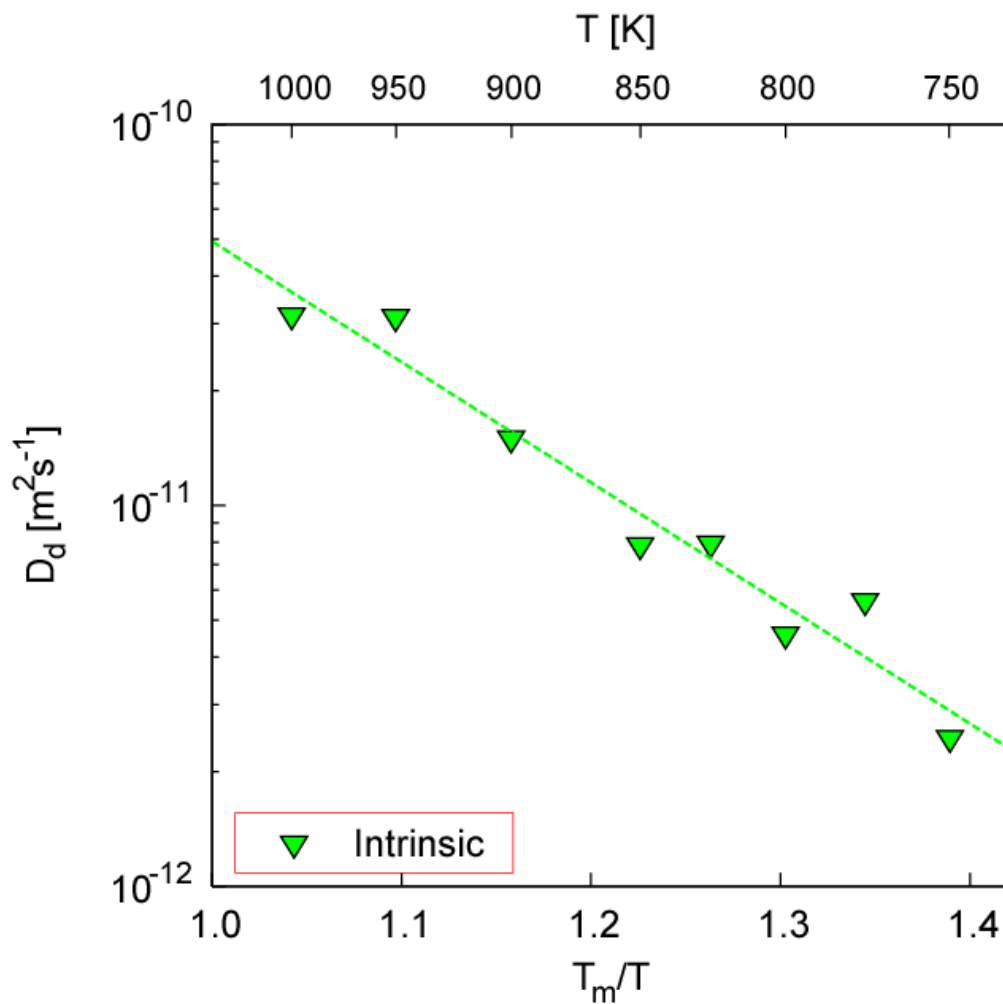
$$D_d = D_d^I + N_v \left( D_d^{\text{raw}} - D_d^I \right)$$

$$N_v = \sum_{i=1}^n \exp\left(-\frac{E_{vi}}{kT}\right)$$

$$D_d^{\text{raw}} \approx A/e; r_d$$

Similar equation for interstitials

# Intrinsic dislocation diffusivity



$$E_d = 0.65 \text{ eV}$$

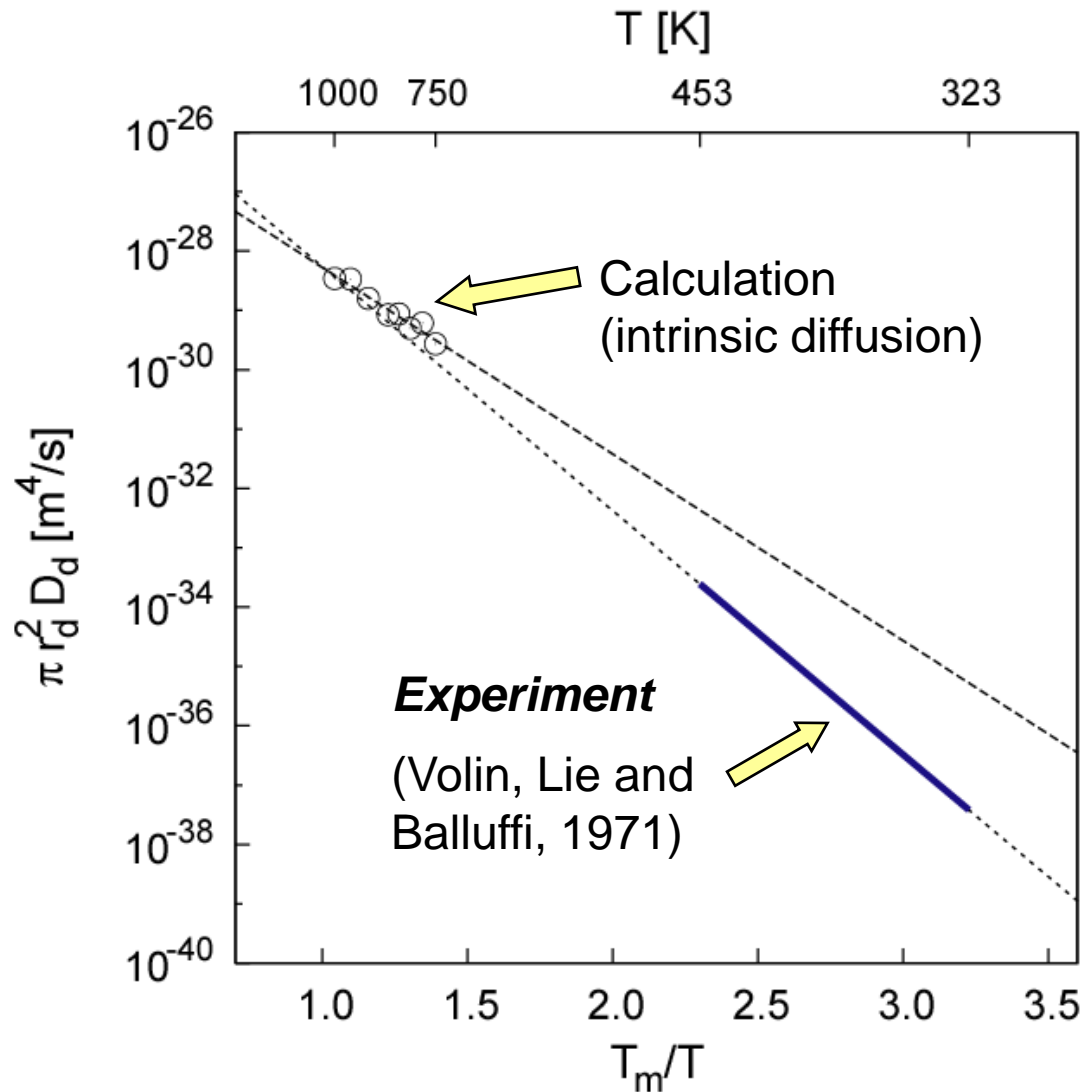
$$D_{d0} = 7.24 \times 10^{-8} \text{ m}^2/\text{s}$$

$$r_d = 0.59 \text{ nm}$$

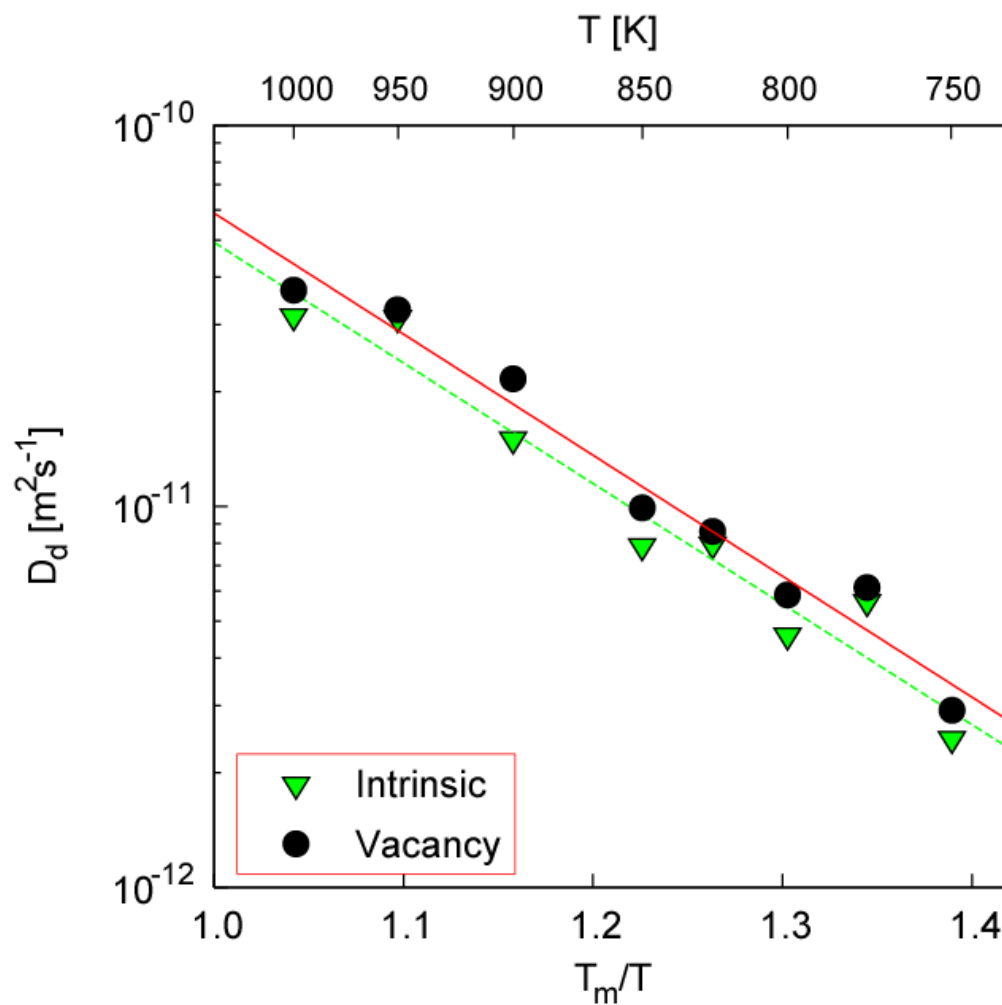
$$E = 1.32 \text{ eV}$$

$$E_d/E = 0.49$$

# Comparison with experiment



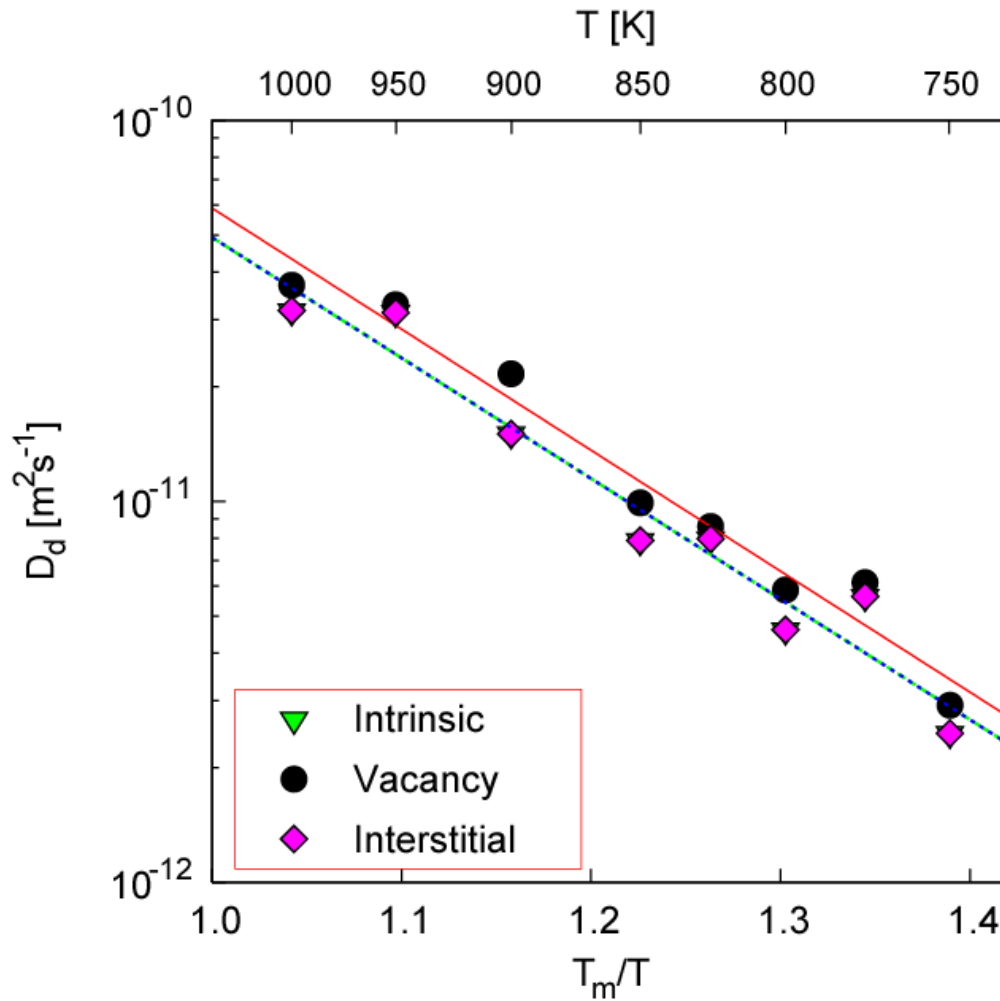
# Contribution of vacancy diffusion



$$E_d = 0.66 \text{ eV}$$

The vacancy contribution is relatively small

# Contribution of interstitial diffusion



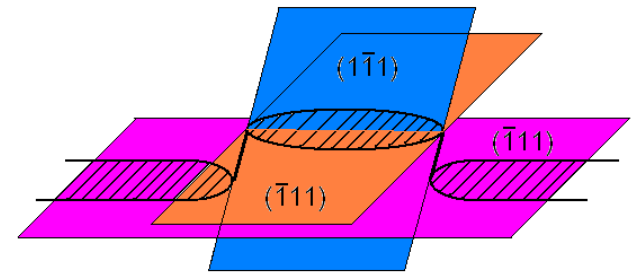
$$E_d = 0.65 \text{ eV}$$

The interstitial contribution is negligible

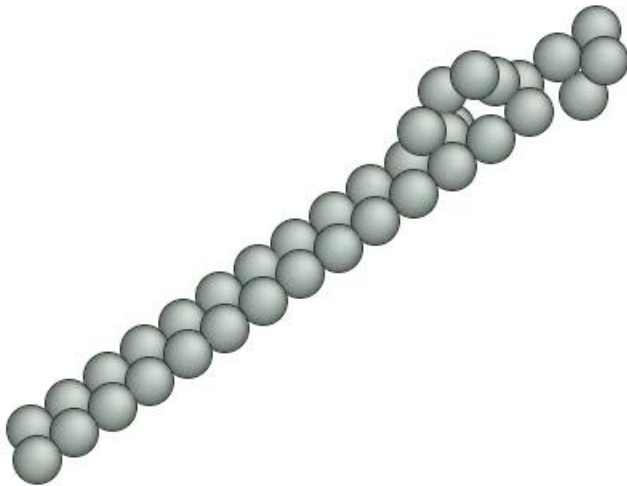
The intrinsic diffusion dominates!

## What is the intrinsic mechanism?

- The dislocation line moves around the average position due to thermal fluctuations
- The motion occurs by the nucleation and spreading of double-jogs  $\Rightarrow$  shuffling of atoms
- This thermal motion has a stochastic component which gives rise to diffusion
- Perfect sliding would translate entire rows  $\Rightarrow$  zero correlation factor
- Need to understand more details



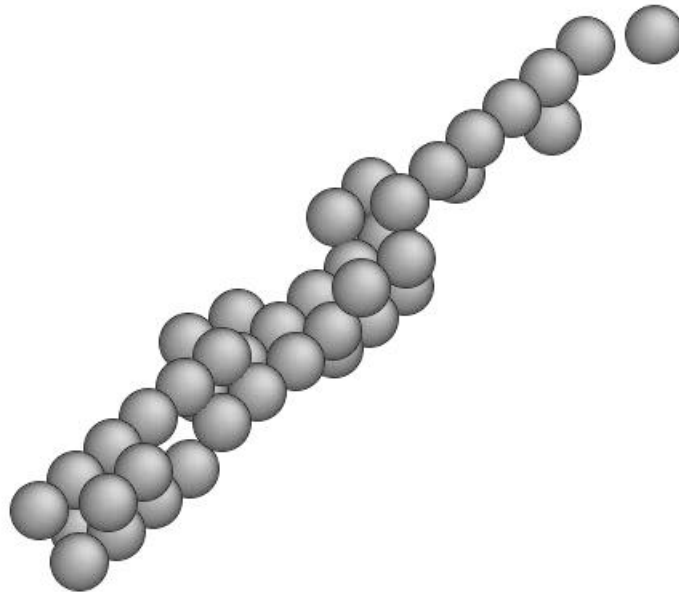
## Dislocation with a vacancy



- The vacancy is wandering around the core
- The vacancy is not absorbed by the core
- Due to the thermal motion, the dislocation easily breaks away from the vacancy
- Does the vacancy induce the jog formation?

Visualization by potential energy

## Dislocation without point defects

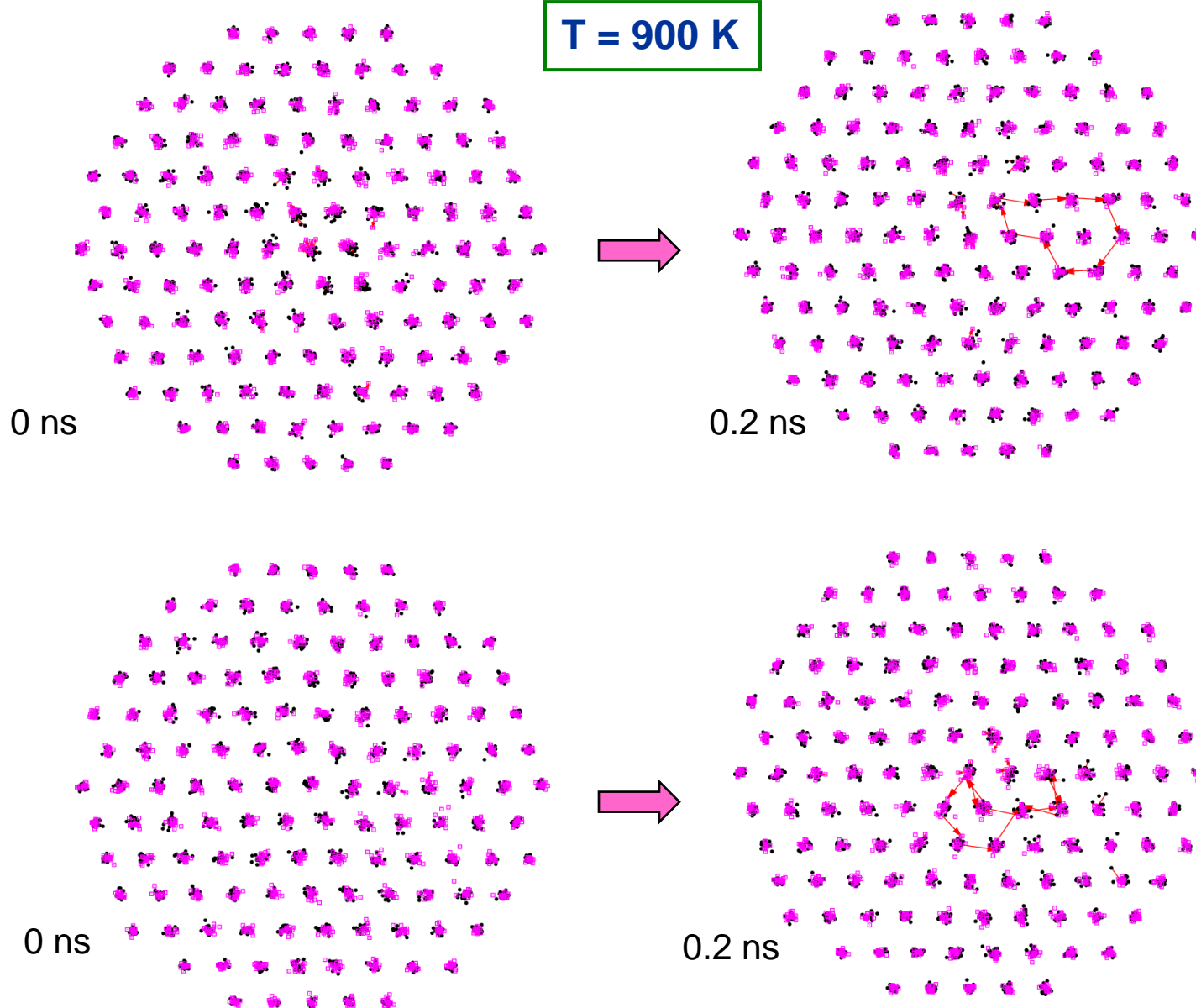


Still observe jog  
formation and thermal  
motion



# Vacancy excursions: Intrinsic case

$T = 900 \text{ K}$



## Ongoing and future work

- What exactly happens in the core during the extrinsic diffusion process?
- Extension to edge and mixed dislocations  
[preliminary result: no significant intrinsic diffusion in edge dislocations]
- Instead of Al, try a metal with a low stacking fault energy
- Extension to the Al-Li
- If the intrinsic diffusion is confirmed, we may need to reconsider the role of point defects in dislocation diffusion